

# Spatial structure of an individual Mn acceptor in GaAs

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The wave function of a hole bound to an individual Mn acceptor in GaAs is spatially mapped by scanning tunneling microscopy at room temperature and an anisotropic, cross-like shape is observed. The spatial structure is compared with that from an envelope-function, effective mass model, and from a tight-binding model. This demonstrates that anisotropy arising from the cubic symmetry of the GaAs crystal produces the cross-like shape for the hole wave-function. Thus the coupling between Mn dopants in GaMnAs mediated by such holes will be highly anisotropic.

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Despite intense study of deep acceptors in III-V semiconductors such as  $\text{Mn}_{\text{Ga}}$ , little information has been obtained on their electronic properties at the atomic scale. Yet the spatial shape of the Mn acceptor state will influence hole-mediated Mn-Mn coupling and thus all of the magnetic properties of hole-mediated ferromagnetic semiconductors [1] such as  $\text{Ga}_{1-x}\text{Mn}_x\text{As}$ . Evidence for anisotropic spatial structure even in shallow acceptors such as Zn, Cd [2, 3] and C [4] in various III-V semiconductors suggest that anisotropic hole states may be common. In addition to controlling inter-dopant properties such as the Mn-Mn coupling [5, 6], the wave-function shape would affect single-dopant properties such as the g-factor and optical transition oscillator strengths.

This study presents an experimental and theoretical description of the spatial symmetry of the Mn acceptor wave-function in GaAs, and we suggest our results imply similar behavior for other acceptors and other hosts. We first present our measurements of the spatial mapping of the anisotropic wave function of a hole localized at a Mn acceptor. To achieve this, we have used the STM tip not only to image the Mn acceptor but also to manipulate its charge state  $A^0/A^-$  at room temperature as described in [7]. Within an envelope function effective mass model (EFM) the anisotropy in the acceptor wave-function can be traced to differing amplitudes of envelope functions with the same total angular momentum ( $L > 0$ ) but different angular momentum projections along a fixed axis. We introduce into the EFM a single parameter  $\eta$  that describes the breaking, by the cubic crystal, of spherical symmetry for the acceptor level envelope functions. As  $\eta$  has a negligible effect on the binding energy compared to the central cell correction, common variational approaches cannot be used to evaluate  $\eta$ . However, comparison with calculations based on a tight-binding model (TBM) for the Mn acceptor structure [6] permits us to clearly identify the physical origin of the anisotropic shape in these models, and to justify the value of  $\eta$  used in our EFM calculations to

describe the experimental shape. The TBM calculations also demonstrate that although the spin-orbit interaction does influence the acceptor wave-function, the qualitative anisotropic shape of the acceptor state occurs in crystals without spin-orbit interaction. Thus acceptor levels in crystals such as GaN should have a similar shape.

The measurements were performed on several samples using chemically etched tungsten tips. The samples consisted of a 1200 nm thick layer of GaAs doped with Mn at  $3 \times 10^{18} \text{ cm}^{-3}$  grown by MBE on an intrinsic (001) GaAs substrate. A growth temperature of 580 °C was chosen to prevent the appearance of structural defects such as As antisites, which would complicate the spatial mapping by shifting the position of the Fermi level of the sample. The concentration of the Mn dopants was low enough to neglect Mn-Mn interactions and the formation of an impurity band. The samples we used were insulating below 77 K. The experiments were performed in a room temperature UHV-STM ( $P < 2 \times 10^{-11}$  torr) on an *in situ* cleavage induced (110) surface.

A major advantage to our approach is that the occupation of the acceptor state can be influenced by band-bending from the voltage applied between the STM tip and the sample (see the Fig. 1 inset). We studied the voltage dependent appearance of the Mn acceptor in the STM constant-current image. In the ionized configuration at high negative bias Mn appeared as an isotropic round elevation which is a consequence of the influence of the  $A^-$  ion Coulomb field on the valence band states [Fig. 2(a)]. This agrees with a recent study of the individual Mn in GaAs in the ionized configuration [8]. We found that at a positive bias Mn is electrically neutral, as can be seen from the absence of the electronic contrast at high positive voltage ( $U > 1.5$  V) when the conduction band empty states dominate tunneling (see the positive branch of the  $I(V)$  curve in Fig. 1). At low positive voltage where the tip Fermi level is below the conduction band edge Mn appeared as a highly anisotropic cross like feature [Figs. 2(b) and 3(a)]. The anisotropy is even more

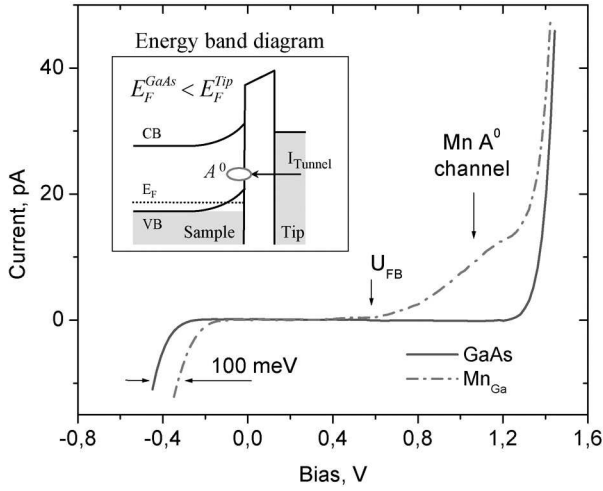


FIG. 1:  $I(V)$  curves acquired on the clean GaAs surface (solid line) and in the middle of the cross-like feature (dashed line). The simulated position of the flat-band potential  $U_{FB}$  is indicated by the arrow. Insert displays energy band diagram for the positive sample-bias.

evident in a reciprocal-space image [Fig. 3(b)]. The methods of calculating the theoretical images [Figs. 2(d), 3(c-f)] will be described below.

The cross like feature manifested itself in the local tunneling  $I(V)$  spectroscopy at low voltages when the GaAs bands do not contribute to the tunneling. It appeared as an empty states or filled states current channel in the band gap of GaAs depending on applied positive or negative bias, respectively. Thus the mapping of the Mn acceptor state in the filled (empty) states mode was realized by electron (hole) injection into the  $A^0/A^-$  state. In the tunneling  $I(V)$  spectroscopy the manganese  $A^0$  channel appeared presumably above the flat band potential  $U_{FB}$  and was available for tunneling in the wide range of voltages above  $U_{FB}$ . Our estimated value of  $U_{FB}$  is about +0.6 V. The observed ionization energy, which was determined from the shift of the  $I(V)$  spectrum at negative bias, corresponds to the Mn acceptor binding energy  $E_a = 0.1$  eV.

The concentration of the dopants we observed with STM corresponds to the intentional  $3 \times 10^{18} \text{ cm}^{-3}$  doping level. All of the dopants could be found either in the ionized  $A^-$  or the neutral  $A^0$  charge state depending on either negative or positive sample bias respectively. In the experiment we identified Mn located in at least 6 different layers under the surface. In order to determine the actual position of the Mn dopants we analyzed the intensity of the electronic contrast of the Mn related features. Based on the symmetry of the cross like feature superimposed on the surface lattice we distinguished whether the dopant is located in an even or odd sub-surface layer. We found that at any depth the shape of the hole on  $\{110\}$  surface had nearly  $C_{2v}$  symmetry around the surface normal, and was elongated along the  $[001]$  direction relative to the  $[1\bar{1}0]$  direction. The cross-like features induced by

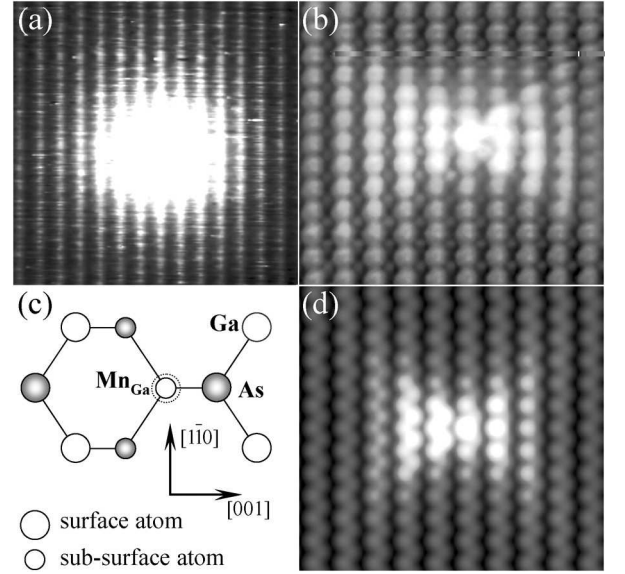


FIG. 2: (a)  $8 \times 8 \text{ nm}^2$  STM image of an ionized Mn acquired at -0.7 V; (b)  $5.6 \times 5 \text{ nm}^2$  STM image of neutral Mn acquired at +0.6 V. Big and small round features correspond to As and Ga related surface states, respectively. Presumably, Mn is located in  $3^{rd}$  sub-surface atomic layer; (c) A model of the  $(110)$ -surface (top view) representing the Mn on Ga site located in an odd sub-surface atomic layer (counting surface layer as zero); (d)  $5.6 \times 5 \text{ nm}^2$  simulated image (logarithm of local density of states) of the Mn located in the  $5^{th}$  sub-surface atomic layer (TBM).

deeper dopants were more elongated in the  $[001]$  direction. The feature was weakly asymmetric with respect to  $[1\bar{1}0]$  direction (lowering the symmetry to the single  $(1\bar{1}0)$  mirror plane) which may come from the symmetry properties of the  $(110)$  surface.

A four-band envelope-function effective Luttinger-Kohn Hamiltonian provided one framework (EFM) in which to analyze the spatial structure of the bulk-like neutral acceptor complex formed by a valence hole loosely bound to a negatively charged  $\text{Mn}^{2+}3d^5$  core ( $\text{Mn}^{2+}3d^5 + \text{hole}$  complex). The ground state of this acceptor in a zincblende semiconductor can be approximated as four-fold degenerate with a total momentum of the valence hole  $F = 3/2$  and has the symmetry of the top of the valence band  $\Gamma_8$  [9]. We neglected possible effects caused by the presence of the  $(110)$  surface and quantum spin effects from the exchange interaction between the  $\text{Mn}3d^5$  core and the hole. We also ignored the excited states, as the energy separation between the ground state  $E_a(1S_{3/2}) = 113 \text{ meV}$  and the first excited state  $E_a(2S_{3/2}) = 25 \text{ meV}$  exceeds room temperature [10].

According to Ref. 9 the acceptor wave-function in zincblende semiconductor is represented as a four-component column written in the basis of Bloch functions of the valence band top  $\Gamma_8$ . The form of the wave-



like shape. However the cross-like shape and the intensity of the satellites are more pronounced in the experimental Fourier spectrum. Based on the analyses we performed we conclude that only the harmonic  $Y_{2,1}$  gives rise to the intensity of the satellites in the Fourier spectrum. The cross feature in the Fourier spectrum is also influenced by the envelope function  $R_2$  and is more pronounced when  $R_2$  has inverse power dependence on distance rather than exponential decay.

The tight-binding model (TBM) we use [6] is based on the deep level model of Vogl and Baranowski [18]. The dangling  $sp^3$ -bonds from the nearest-neighbor As hybridize with the Mn d-states of  $\Gamma_{15}$  character. The antibonding combination of these becomes the Mn acceptor state. Coupling to the d-states of  $\Gamma_{12}$  character is weak, and hence neglected. The hybridization strength is fully determined by the acceptor level energy. This model, if further approximated within the EFM, predicts  $\eta = 0$ , similar to what we found by fitting the EFM to the experimental measurements.

The calculations of the local density of states (LDOS) based on the TBM are shown in logarithmic scale in Fig. 2(d) and Figs. 3(e,f). The results show symmetry under reflection in the  $(1\bar{1}0)$  plane, and asymmetry under reflection in the  $(001)$  plane. For Mn dopants several layers down from the surface, as in Fig. 2(b), the shape of the acceptor state does not depend that sensitively on the spin orientation of the Mn-core  $3d$ -spin. For these dopants the shape does not depend on the spin-orbit interaction; we confirmed this by obtaining a similar cross-like acceptor structure using a tight-binding Hamiltonian without spin-orbit interaction, and with empirical parameters designed for optimal agreement with the bulk band structure of GaAs [19]. The situation differs greatly for Mn near the surface, where the axis of extension of the acceptor state rotates with the spin orientation of

the Mn core spin. At these temperatures we can expect the Mn core spin to point in a random direction, and for the STM measurements to average over the possible spin orientations. Thus for Figs. 2(d) and 3(e,f) the LDOS is averaged over the Mn spin orientation.

Although the symmetry is well retained for all simulated positions of Mn under the surface, in both models the best fits were achieved when the apparent depth of the Mn is assumed to be 2 atomic layers larger. The reason for this could be the vacuum barrier, which will tend to shift the wave function of the Mn acceptor state deeper into the crystal than one would predict for the sliced bulk crystal calculation. The lateral size of the wave function measured is also somewhat larger than calculated for Fig. 3; this might come from a reduction of the acceptor binding energy very near the surface.

In conclusion, we have experimentally demonstrated that the Mn acceptor ground state has highly anisotropic spatial structure. This spatial anisotropy is due to a significant presence of d-wave envelope functions in the acceptor ground state. We have demonstrated that the observed symmetry can be explained within simple tight-binding model whose only free parameter is the acceptor level energy. We also found that this spatial structure can be described well by a simple four-band envelope-function model of cubic symmetry whose key parameter,  $\eta$ , can be fit to the observed spatial structure, and is similar to that expected from a tight-binding model. These results have broad implications for all acceptor-acceptor interactions in zincblende semiconductors, and especially for hole-mediated ferromagnetic semiconductors.

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